# A new polymorph of LiZnPO<sub>4</sub>·H<sub>2</sub>O; synthesis, crystal structure and thermal transformation †

DALTON FULL PAPER

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A new polymorph of lithium zinc phosphate hydrate,  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O, has been obtained by hydrothermal synthesis, and its crystal structure and thermal transformation investigated. The crystal structure can be viewed as a framework structure built from PO<sub>4</sub> and ZnO<sub>4</sub> tetrahedra with LiO<sub>4</sub> tetrahedra and water molecules placed in eight-ring channels;  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O is apparently the only member of the zeolite ABW group of materials crystallising in space group  $P2_1ab$ . It has a doubled b axis compared to a known phase of LiZnPO<sub>4</sub>·H<sub>2</sub>O. The <sup>31</sup>P MAS NMR spectrum showed two signals;  $\delta$  6.94(2) and 7.88(2). The known phase of LiZnPO<sub>4</sub>·H<sub>2</sub>O transforms into  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O after treatment in 0.9 M LiNO<sub>3</sub> (aq) at 80 °C and 10 kbar for 20 h. A mechanism for the pressure-induced displacive phase transition is proposed. Dry or hydrothermal heating of  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O at >100 °C gave  $\delta$ <sub>1</sub>-LiZnPO<sub>4</sub>. The enthalpy change for the dehydration with a maximum at 219(2) °C was determined to be  $\Delta H = 43(6)$  kJ mol<sup>-1</sup>, using differential scanning calorimetry.

Microporous materials have found extensive industrial use over the last decades, *e.g.* as gas absorbers, ion exchangers, catalysts and many other applications. Such materials with new framework structures and compositions are of great importance as their function is often correlated with the internal shape and size of cages and channels and the position of metal ions in the material. Among new materials a large number of phosphates and some arsenates have been characterised during the last decade.<sup>2</sup>

The chemistry of zinc orthophosphates shows a large variety of phases mainly due to the co-ordination flexibility of the zinc ions and their ability to form Zn-O-Zn linkages.<sup>3</sup> An increasing number of orthophosphates are found in the system, ZnO-P<sub>2</sub>O<sub>5</sub>-cation, prepared by hydrothermal synthesis from aqueous or non-aqueous solutions using alkali-metal ions and/or organic amine ions as templates. There are several members of this group which are structural analogues to aluminosilicates, e.g. sodalite, zeolite-X and ABW,4 but also compounds with novel crystal structures with no naturally occurring mineral counter part, e.g. Zn<sub>2</sub>(HPO<sub>4</sub>)<sub>3</sub>·H<sub>3</sub>NCH<sub>2</sub>CH<sub>2</sub>NH<sub>3</sub>.<sup>5</sup> A member of this family, Zn(H<sub>2</sub>PO<sub>4</sub>)(HPO<sub>4</sub>)·(CH<sub>3</sub>)<sub>4</sub>N, possesses very low framework density, 10.1 tetrahedral framework atoms (T) per 1000 Å<sup>3.6</sup> In general microporous materials are metastable as they transform to more dense phases upon heating either dry or hydrothermal. Careful adjustment of synthesis conditions is crucial for preparation of new materials, e.g. stabilisation and isolation of intermediate phases in the synthesis of more dense

Hydrothermal preparation techniques have revealed the existence of several new lithium zinc phosphates and the crystallisation fields of the compounds are under further investigation. A polymorph of LiZnPO<sub>4</sub> having a phenacite type structure (denoted  $\epsilon$ -LiZnPO<sub>4</sub> in this work for distinction from the other polymorphs), a cristobalite related phase,  $\delta_{\rm I}$ -LiZnPO<sub>4</sub>, and a polymorph (denoted  $\alpha'$ -LiZnPO<sub>4</sub>) with a more complex crystal structure can all be prepared by hydrothermal methods. The compound  $\alpha'$ -LiZnPO<sub>4</sub> is possibly structurally related to  $\alpha$ -LiZnPO<sub>4</sub>, which is the stable phase under ambient conditions prepared by solid state reaction. A common feature of the lithium zinc phosphates,  $\alpha$ -,  $\epsilon$ - and  $\delta_{\rm I}$ -LiZnPO<sub>4</sub> (and possibly also  $\alpha'$ -LiZnPO<sub>4</sub>), are three-dimensional framework structures built from LiO<sub>4</sub>, ZnO<sub>4</sub> and PO<sub>4</sub> tetrahedra forming

six-ring channels. The system Li<sub>3</sub>PO<sub>4</sub>–Zn<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub> was studied by means of solid state reactions and thermal analysis. The equilibrium phase diagram shows the existence of four lithium zinc phosphates with the compositions Li<sub>4</sub>Zn(PO<sub>4</sub>)<sub>2</sub>, Li<sub>9</sub>-Zn<sub>6</sub>(PO<sub>4</sub>)<sub>7</sub>, LiZnPO<sub>4</sub> and 'LiZn<sub>9</sub>(PO<sub>4</sub>)<sub>7</sub>', and each of them may exist in one, two or three modifications. The phase diagram reveals that  $\alpha$ -LiZnPO<sub>4</sub> transforms to  $\beta$ -LiZnPO<sub>4</sub> at 737 °C,  $\gamma$ -LiZnPO<sub>4</sub> at 1006 °C and melts at 1172 °C. It is not possible to quench  $\beta$ - or  $\gamma$ -LiZnPO<sub>4</sub> to ambient conditions as the phase transitions are reversible and the only available structural data is a powder pattern of  $\beta$ -LiZnPO<sub>4</sub>. A polymorph of Li<sub>4</sub>-Zn(PO<sub>4</sub>)<sub>2</sub> and the mentioned phases of LiZnPO<sub>4</sub> seem to be the only compositions from the phase diagram that can be prepared by means of hydrothermal synthesis. I<sup>2</sup>

A water-containing lithium zinc phosphate, LiZnPO<sub>4</sub>·H<sub>2</sub>O, was described having a zeolite type ABW structure, *i.e.* isomorphous with zeolite Li-A(BW), LiAlSiO<sub>4</sub>·H<sub>2</sub>O. <sup>9a</sup> An accurate structure analysis of LiAlSiO<sub>4</sub>·H<sub>2</sub>O was performed by combining single crystal and powder diffraction data, using X-rays and neutron radiation, respectively. The structure can be viewed as a framework built from AlO<sub>4</sub> and SiO<sub>4</sub> tetrahedra giving four- and eight-ring channels in the crystallographic c direction and six-ring channels in a perpendicular direction. The lithium ions are co-ordinated to three framework oxygen atoms and one water oxygen atom. <sup>13</sup>

The polymorph of LiZnPO<sub>4</sub>·H<sub>2</sub>O <sup>9a</sup> found to be isomorphous with zeolite ABW is denoted  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O in the following. The polymorph of LiZnPO<sub>4</sub>·H<sub>2</sub>O discovered in this study has similar unit cell parameters to those of  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O, but the *b* axis is doubled, therefore, denoted  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O. The synthesis, crystal structure and thermal properties of  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O are now reported.

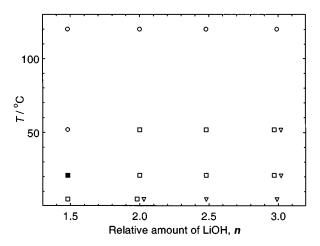
# **Experimental**

## Hydrothermal synthesis

In a search for new materials the system  $\text{Li}_2\text{O}-\text{ZnO}-\text{P}_2\text{O}_5-\text{template}$  was investigated by hydrothermal synthesis in aqueous and non-aqueous solutions using organic amine ions as templates. A synthesis was performed by dissolving  $\text{LiOH}\cdot\text{H}_2\text{O}$  (3.677 g), LiCl (3.709 g) and  $\text{Zn}(\text{CH}_3\text{CO}_2)_2\cdot\text{2H}_2\text{O}$  (18.971 g) in water (100 cm³). Phosphoric acid, 85%  $\text{H}_3\text{PO}_4$  (10.0 cm³), and then ethyldiisopropylamine (30.0 cm³) was slowly added with stirring giving a molar ratio of the reactants  $\text{Li}:\text{Zn}:\text{PO}_4:$  NEtPr $^1_2:$  water of 1.2:0.6:1:1.2: $\approx$ 41,  $[\text{PO}_4]\approx$ 0.9 M, pH 4.55

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<sup>†</sup> Non-SI unit employed: bar = 101 325 Pa.



**Fig. 1** The crystallisation fields of lithium zinc orthophosphates formed in the system LiOH–Zn(CH<sub>3</sub>CO<sub>2</sub>)<sub>2</sub>–H<sub>3</sub>PO<sub>4</sub>-water, at varying amounts of LiOH, n, and temperature. In one case a phase-pure sample of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O (■) was obtained. Open symbols indicate impurities of β-Li<sub>3</sub>PO<sub>4</sub>, and α-LiZnPO<sub>4</sub>·H<sub>2</sub>O is shown as triangles, β-LiZnPO<sub>4</sub>·H<sub>2</sub>O as squares and α'-LiZnPO<sub>4</sub> as circles

(measured 1 h after mixing). After stirring the gel for 1 h, powder diffraction showed the presence of  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O of poor crystallinity along with a trace of an unidentified phase. A portion of the gel, ca. 5 cm<sup>3</sup>, was left to stand for 3 months at room temperature, ca. 21 °C, before crystallisation was noticed and after 6 months the crystallisation was complete and the material (Sample I) was recovered by filtration and washed with deionised water. Another portion of the gel was heated at 55 °C for 3 months giving  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O as product. The rest of the gel was heated to 200 °C for 68 h in a Teflon-lined steel autoclave giving  $\delta_1$ -LiZnPO<sub>4</sub> as product.

In order to investigate the crystallisation fields of lithium zinc phosphates formed in the system LiOH–Zn(CH<sub>3</sub>CO<sub>2</sub>)<sub>2</sub>–H<sub>3</sub>PO<sub>4</sub>– water, a number of syntheses were performed varying the amount of LiOH and the temperature. The LiOH and Zn(CH<sub>3</sub>CO<sub>2</sub>)<sub>2</sub> were dissolved in water and H<sub>3</sub>PO<sub>4</sub> was slowly added with stirring forming a white gel. The relative amount of lithium is given as a variable n, relative to the amount of phosphoric acid:  $n = n(\text{Li})/n(\text{PO}_4)$ . The molar composition of the synthesis mixtures, Li: Zn: PO<sub>4</sub>: H<sub>2</sub>O, was  $n: 0.52: 1: \approx 20$ , where  $n = 1.49, 2.00, 2.48 \text{ or } 2.99, [PO_4] \approx 1.8 \text{ m}, 4.0 < pH < 7.2, ther$ mal treatment at 5, 21 or 52 °C for 28 d and at 120 °C for 6 d. The phase identification was performed by combining information from 31P MAS NMR and powder diffraction and the products are displayed in Fig. 1. In one case a phase-pure sample of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O was obtained (filled square) and the other samples contained impurities of β-Li<sub>3</sub>PO<sub>4</sub> (open symbols). At high temperatures, e.g. 120 °C, α'-LiZnPO<sub>4</sub> (circles) formed and at low temperatures α-LiZnPO<sub>4</sub>·H<sub>2</sub>O (triangles) was found as product. Optimisation of the conditions for the hydrothermal synthesis revealed that templates were in fact not essential for the preparation of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O. Several syntheses in the system LiOH-LiCl-Zn(CH<sub>3</sub>CO<sub>2</sub>)<sub>2</sub>-H<sub>3</sub>PO<sub>4</sub>-(CH<sub>2</sub>OH)<sub>2</sub> were performed in a similar way substituting the solvent water with ethylene glycol but gave only semiamorphous material with traces of  $\alpha'$ -LiZnPO<sub>4</sub> and  $\beta$ -Li<sub>3</sub>PO<sub>4</sub> as products.

A Teflon container holding a suspension of  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O (0.381 g) in 0.9 m LiNO<sub>3</sub> (aq) (10 cm³) and H<sub>3</sub>PO<sub>4</sub> (0.1 cm³) were heated in a PSIKA 20 kbar reactor at 80 °C and 10 kbar for 20 h, using oil as supporting fluid. The composition of the suspension was Li:Zn:PO<sub>4</sub>:water = 2.6:0.58:1: $\approx$ 142. The product was washed with ethanol (99.9%) and identified as  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O using powder diffraction.

A Radiometer glass electrode combined with a calomel reference electrode (GK2401C) was used for pH measurements. The syntheses were performed using the following commercial chemicals: LiOH·H<sub>2</sub>O (Fluka, puriss p.a., >99%), LiNO<sub>3</sub>

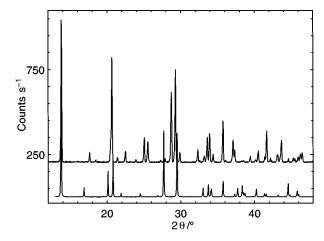


Fig. 2 Powder X-ray diffraction patterns of  $\alpha\text{-}$  (bottom) and  $\beta\text{-LiZnPO}_4\text{-}H_2O$ 

(Merck, >98%), LiCl (Merck, pro analysi, >99%), H<sub>3</sub>PO<sub>4</sub> (85%, Fluka, extra pure), Zn(CH<sub>3</sub>CO<sub>2</sub>)<sub>2</sub>·2H<sub>2</sub>O (Fluka, purum p.a. >99.0%), ethyldiisopropylamine (Aldrich, >99%) and ethylene glycol (Fluka, puriss p.a., >99.5%).

#### **Powder diffraction**

Powder diffraction data for phase identification and refinement of unit cell parameters were obtained using a Siemens D5000 diffractometer equipped with a primary germanium monochromator (Cu-K $\alpha_1$  radiation,  $\lambda = 1.540598$  Å). The data were collected from 5 and up to 90° in 20 with a step length of 0.02° and counting time of 2-15 s per step. The observed powder pattern of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O was distinctly different from that of the known zeolite type ABW polymorph, α-LiZnPO<sub>4</sub>·H<sub>2</sub>O, as shown in Fig. 2. The trial and error indexing program TREOR 14 revealed an orthorhombic unit cell of β-LiZn-PO<sub>4</sub>·H<sub>2</sub>O similar to the unit cell of α-LiZnPO<sub>4</sub>·H<sub>2</sub>O with a doubled b axis. The program CELLKANT 15 was used to refine the unit cell parameters from the observed d spacings. The indexed powder pattern of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O (39 reflections,  $I/I_0 > 2\%$ ) is given in Table 1 using the refined unit cell dimensions of  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O, a = 10.030(1), b = 16.553(2) and c = 5.0120(5) Å, unit cell volume = 832.1(2) Å<sup>3</sup>.

#### Single crystal diffraction

A clear, colourless octahedron of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O was selected from sample I. The data collection was performed on a Siemens SMART diffractometer equipped with a CCD detector and graphite monochromatised Mo-Kα radiation,  $\lambda$  = 0.710 73 Å. Approximately one hemisphere of data was collected in frames covering 0.3° in  $\omega$  in three sets at different  $\varphi$  angles using a detector to crystal distance of 40.0 mm. Data were corrected for Lorentz-polarisation effects and for absorption using Gaussian integration [transmission (maximum/minimum) 0.6383/0.3748] giving 1359 unique reflections.

A structural model was found using direct methods in space group  $P2_1ab$ , no. 29. This setting was chosen to resemble the unit cell of  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O refined in space group  $Pna2_1$ , no. 33. Programs from the Siemens SHELXTL M software package were used for structure solution and refinement. After anisotropic refinement it was possible to locate the positions of the four hydrogen atoms in the Fourier-difference maps. The H atom positions and thermal parameters were not refined; all other atoms were refined with anisotropic thermal parameters. Scattering factors of neutral atoms were applied throughout. The refinement converged at R = 0.0290 and R' = 0.0717, with 146 parameters refined using full-matrix least squares based on all  $F^2$ . Atomic coordinates are shown in Table 2. Drawings of the crystal structure were prepared using the program ATOMS. 17

Table 1 Indexed powder pattern of β-LiZnPO <sub>4</sub> · $H_2O$						
h	k	1	$d_{ m calc}$ /Å	$d_{ m obs}$ /Å	$(I/I_0)_{\rm calc}$	$(I/I_0)_{\rm o}$
1	2	0	6.3838	6.3834	100	100
2	0	0	5.0152	5.0173	4	7
1	1	1	4.3275	4.3299	12	7
0	2	1	4.2872	4.2842	79	74
0	4	0	4.1382	4.1438	3	4
1	2	1	3.9422	3.9433	6	6
2	0	1	3.5452	3.5451	22	18
1	3	1	3.4796	3.4827	16	15
2	4	0	3.1919	3.1938	2	3
3	2	0	3.1001	3.1009	58	49
1	4	1	3.0409	3.0426	69	65
2	3	1	2.9826	2.9826	8	7
0	5	1	2.7623	2.7644	9	9
2	4	1	2.6923	2.6918	5	4
1	5	1	2.6632	2.6629	16	18
3	2	1	2.6365	2.6352	24	21
3	4	0	2.6007	2.6012	7	6
0	0	2	2.5060	2.5074	32	30
2	5	1	2.4196	2.4194	16	16
1	1	2	2.4055	2.4052	8	7
0	3	2 2 2	2.2817	2.2818	5	4
2	1	2	2.2214	2.2229	8	9
2	2		2.1645	2.1649	22	22
0	7	1	2.1386	2.1384	2	3
1	7	1	2.0916	2.0916	8	6
0	8	0	2.0691	2.0688	14	15
4	4	1	1.9717	1.9712	3	4
3	6	1	1.9587	1.9593	5	6
3	2	2	1.9489	1.9498	6	7
4	6	0	1.8562	1.8562	7	8
2	8	1	1.7870	1.7869	6	5
4	6	1	1.7402	1.7399	8	8 5
4	3	2	1.6876	1.6880	5	5
5	6	0	1.6225	1.6224	7	7
0	8	2	1.5955	1.5957	6	7
6	0	1	1.5858	1.5856	9	7
1	3	3	1.5790	1.5784	5	8

**Crystal data.** LiO<sub>4</sub>PZn·H<sub>2</sub>O, M = 185.30, orthorhombic, space group  $P2_1ab$  (no. 29), a = 10.0224(9), b = 16.5596(15), c = 5.0126(5) Å, U = 831.93(13) Å<sup>3</sup>,  $D_c = 2.959$ ,  $D_m = 3.01$  g cm<sup>-3</sup> (measured using a pycnometer and water), T = 293 K, Z = 8,  $\mu$ (Mo-K $\alpha$ ) = 6.189 mm<sup>-1</sup>, 3348 reflections measured, 1359 unique ( $R_{\rm int} = 0.0362$ ).

1.5720

1.5627

13

11

5

CCDC reference number 186/1004.

1.5719

1.5627

See http://www.rsc.org/suppdata/dt/1998/2261/ for crystallographic files in .cif format.

## Spectroscopic investigation

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Magic angle spinning <sup>31</sup>P NMR spectra were recorded on a Varian UNITY-500 spectrometer [11.7 T,  $v(^{31}P) = 202.332$  MHz], using a Jakobsen MAS probe. <sup>18</sup> The samples were placed in a cylindrical rotor (Si<sub>3</sub>N<sub>4</sub>, 5 mm outside diameter, volume 220 µl). The free induction decays were recorded with a spinning speed of 5.5 kHz and the resulting spectra had a digital resolution of 2.5 Hz. An aqueous solution of H<sub>3</sub>PO<sub>4</sub> (85%, Fluka, extra pure) was used as external standard. Infrared spectra were measured on a Perkin-Elmer 1720 Fourier-transform spectrometer using pellets of KBr containing 1% of the compound. The spectral resolution was 4 cm<sup>-1</sup>.

## Thermal investigation

A thermogravimetric (TG) measurement was performed between 30 and 500 °C, using a SETARAM TG 92-12 instrument, a nitrogen atmosphere and a heating rate of 5 °C  $\min^{-1}$ . The sample was kept in an open  $Al_2O_3$  crucible. Differential scanning calorimetry (DSC) was performed using a

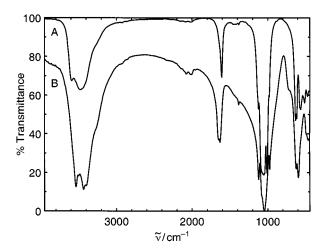


Fig. 3 Infrared spectra of α- (A) and β-LiZnPO<sub>4</sub>·H<sub>2</sub>O (B)

SETARAM DTA 92-16.18 instrument. The sample was placed in a platinum crucible and  $\alpha\text{-Al}_2O_3$  used as reference. The experiment was carried out between 20 and 900 °C using a heating and cooling rate of 5 °C min $^{-1}$  in an argon atmosphere. Two experiments were carried out using the same sample. The temperature and enthalpy change calibration was carried out using the low–high quartz transition and the dehydration of CaSO<sub>4</sub>·2H<sub>2</sub>O. Baseline fluctuations in the temperature range 20 to 130 °C are due to small differences between the set temperature and the actual temperature in the DSC instrument.

# **Results and Discussion**

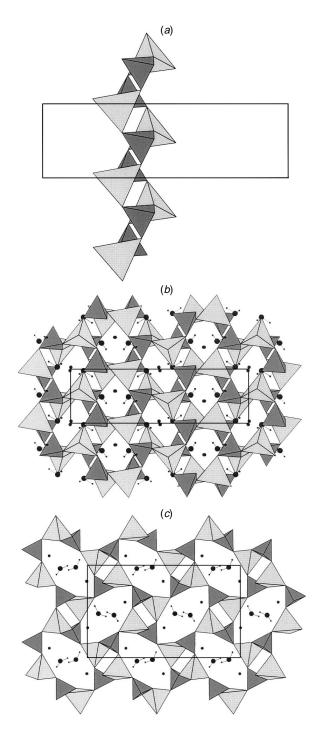
Thermogravimetric measurements were in accordance with the formula  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O, observed weight loss 9.1% at 120–210 °C (calculated 9.7%). The density of  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O was measured using pycnometry and water to  $D_{\rm m}=3.01~{\rm g~cm^{-3}}$ , and the calculated density was  $D_{\rm c}=2.959~{\rm g~cm^{-3}}$ . The density of  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O is  $D_{\rm c}=2.886~{\rm g~cm^{-3}}$ .

Phosphorus-31 MAS NMR spectroscopy on β-LiZn-PO<sub>4</sub>·H<sub>2</sub>O showed two signals with chemical shifts  $\delta(^{31}P)$  6.94(2) and 7.88(2), in accordance with the crystallographic data revealing two phosphorus atoms in the asymmetric unit. The two polymorphs of LiZnPO<sub>4</sub>·H<sub>2</sub>O thus have distinct phosphorus chemical shifts as α-LiZnPO<sub>4</sub>·H<sub>2</sub>O shows only one signal [at  $\delta(^{31}P)$  6.05(2)].

Infrared spectra of the two polymorphs of LiZnPO<sub>4</sub>·H<sub>2</sub>O are illustrated in Fig. 3. The absorptions at *ca.* 3500 and *ca.* 1600 cm<sup>-1</sup> are due to stretch and bending vibrations in water molecules. A broad absorption in the spectrum of α-LiZnPO<sub>4</sub>·H<sub>2</sub>O around 3400 cm<sup>-1</sup> indicates the presence of adsorbed water, whereas multiple absorptions in the range 3300–3600 cm<sup>-1</sup> suggest more than one inequivalent crystal water molecule in β-LiZnPO<sub>4</sub>·H<sub>2</sub>O. Orthophosphates generally show a strong absorption band in the range 940–1120 and medium at 540–650 cm<sup>-1</sup>, due to framework vibrations.<sup>19</sup> Differences in this part of the spectra, Fig. 3, suggest differences in the framework structure of the two polymorphs, as will be discussed in the following.

#### Crystal structure of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O

The crystal structure of  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O is built from regular tetrahedra of ZnO<sub>4</sub> and PO<sub>4</sub> and slightly distorted tetrahedra of LiO<sub>4</sub>. Selected bond lengths and angles are found in Table 3. Chains of alternating ZnO<sub>4</sub> and PO<sub>4</sub> tetrahedra run in the c direction. One half the chains have all tetrahedra pointing up (U) and the other chains have tetrahedra pointing down (D). Fig. 4(a) illustrates the connection between two chains forming zigzag four-ring chains parallel to the c axis. Fig. 4(b) displays how the double chains of four-ring chains are interconnected to



**Fig. 4** Crystal structure of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O: (a) the b,c projection, showing four-ring chains in the c direction; (b) connectivity of the four-ring chains forming nets of six-ring channels with stacking sequence, ABA, in the (100) direction; (c) the a,b projection, showing four- and eight-ring channels in the c direction with water and Li atoms placed in the eight rings. The ZnO<sub>4</sub> are grey and PO<sub>4</sub> are dark tetrahedra, lithium ions are shown as medium size black circles and water molecules as small (H atoms) and large (O atoms) black circles

form nets of six rings with the tetrahedra sequence UUUDDD, a characteristic feature of the ABW topology.  $^{6b,13}$  Two types of six rings appear trigonally and ovally distorted having the symmetry of twofold axes and a glide planes, respectively. The trigonal shape resembles the shape of six rings found in the structures of  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O and tridymite, whereas the oval resembles  $\delta_1$ -LiZnPO<sub>4</sub> and cristobalite (the sequence of tetrahedra in a six ring of the three latter compounds are UDUDUD).  $^{9b,20}$  The b,c projection, Fig. 4(b), consists of pseudo-hexagonal layers stacked with the sequence ABA to

**Table 2** Atomic coordinates (×10<sup>4</sup>) and equivalent isotropic displacement parameters (Å<sup>2</sup> × 10<sup>3</sup>) for β-LiZnPO<sub>4</sub>·H<sub>2</sub>O;  $U_{\rm eq}$  is defined as one third of the trace of the orthogonalised  $U_{ij}$  tensor. Parameters for H atoms were not refined

Atom	X	y	Z	$U_{ m eq}$
Zn(1)	1521(1)	6912(1)	4606(2)	10(1)
Zn(2)	9184(1)	5564(1)	1108(2)	10(1)
P(1)	8494(2)	6581(1)	5875(3)	7(1)
P(2)	7232(2)	4082(1)	149(4)	8(1)
Li(1)	9007(17)	7524(8)	923(20)	16(3)
Li(2)	1768(13)	4971(7)	4694(24)	15(3)
O(1)	9584(5)	7041(3)	4330(10)	13(1)
O(2)	1915(5)	5974(3)	6867(9)	13(1)
O(3)	2393(5)	6764(3)	1078(9)	11(1)
O(4)	8326(5)	5740(3)	4615(10)	10(1)
O(5)	1116(5)	5463(3)	1360(10)	12(1)
O(6)	8865(5)	6510(3)	8823(9)	12(1)
O(7)	2181(5)	7953(3)	5676(10)	14(1)
O(8)	8531(5)	4552(3)	9731(10)	12(1)
O(9)	393(6)	8247(3)	9846(11)	22(1)
O(10)	261(6)	4260(4)	5348(11)	23(1)
H(1)	1026	8000	8687	90
H(2)	84	8707	9102	90
H(3)	529	3788	6369	90
H(4)	9428	4500	5914	90

form a three-dimensional framework structure, as illustrated in Fig. 4(c) viewed along the c axis, showing another characteristic feature of the ABW topology: four- and eight-ring channels in the (001) direction. The water molecules and lithium ions are placed in the eight-ring channels as shown in Fig. 4(c).

Oxidation states,  $V_i$ , were calculated as a sum of bond valences, s, using the equation  $s = \exp[(r_0 - r)/B]$ , where  $r_0$  and B are empirical parameters  $[r_0(P-O) = 1.617, r_0(Zn-O) = 1.704,$  $r_0(\text{Li-O}) = 1.466$ ,  $r_0(\text{H-O}) = 0.882$  and B = 0.37 Å] and r is the bond length from the refined structural model.<sup>21</sup> The calculated oxidation states,  $V_i$ , are in Table 4, and the calculated values are in agreement with the expected oxidation states. The values of the oxygen atoms fall in three groups. The framework oxygen atoms co-ordinated to one P, Zn and Li each, O(1)-O(6), have calculated oxidation states,  $V_{\rm i}$ , in the range 1.94–2.01. The framework oxygen atoms co-ordinated to two cations, O(7) and O(8), have lower  $V_i$ , 1.83 and 1.81, possibly due to a bond valence contribution from hydrogen bonding. The oxygen atoms in the water molecules, O(9) and O(10), gave values of 2.07 and 1.75, respectively. The low value of O(10) is due to long O(10)–H(3,4) bonds.

# Comparison with related zeolite ABW type materials

A variety of different materials adopt the zeolite ABW type structure and crystallise with at least six different space group symmetries. A new modification of the ABW structure is adopted by  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O crystallising with space group symmetry  $P2_1ab$ . The material LiAlSiO<sub>4</sub>·H<sub>2</sub>O, zeolite Li-A(BW), and  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O crystallise in the space group  $Pna2_1$ , i.e. the twofold screw axes are placed inside and parallel to the four-ring chains. The tetrahedra in each four ring are pairwise symmetry related and all tetrahedra point in the same direction viewed along the polar c axis in the structure of  $\alpha$ - (see Fig. 5) whereas in  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O one half point up and one half down, UUDD [see Fig. 4(c)]. In the structure of  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O the twofold screw axes are in the centre of the six rings, and perpendicular to the four-ring chains, i.e. the tetrahedra in a four ring are crystallographically distinct.

The connectivity of the tetrahedra is the same for LiAl-SiO<sub>4</sub>·H<sub>2</sub>O,  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O and  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O; they all have the characteristic ABW topology and as other lithium-containing zeolitic materials they have high framework densities, 19.0, 18.8 and 19.2 T/1000 ų, respectively.

**Table 3** Selected bond lengths (Å) and angles (°) for β-LiZnPO<sub>4</sub>·H<sub>2</sub>O. Symmetry transformations used to generate equivalent atoms: (*i*)  $x-1,y,z;(ii)\,x,y,z-1;(iii)\,x+1,y,z;(iv)\,x+\frac{1}{2},-y+\frac{3}{2},z;(v)\,x+\frac{1}{2},-y+1,-z+1;(vi)\,x+\frac{1}{2},-y+1,-z;(vii)\,x+1,\,y,\,z-1;(viii)\,x-\frac{1}{2},-y+1,-z+1;(ix)\,x-1,y,z+1$ 

Zn(1)–O(7) Zn(1)–O(1 <sup>i</sup> ) Zn(1)–O(2) Zn(1)–O(3)	1.922(5) 1.958(5) 1.964(5) 1.988(5)	$\begin{array}{l} O(7)-Zn(1)-O(1^i) \\ O(7)-Zn(1)-O(2) \\ O(1^i)-Zn(1)-O(2) \\ O(7)-Zn(1)-O(3) \\ O(1^i)-Zn(1)-O(3) \\ O(2)-Zn(1)-O(3) \end{array}$	105.3(2) 118.6(2) 109.1(2) 102.0(2) 112.7(2) 109.1(2)
$Zn(2)$ – $O(8^{ii})$ $Zn(2)$ – $O(5^{iii})$ $Zn(2)$ – $O(6^{ii})$ Zn(2)– $O(4)$	1.927(5) 1.948(5) 1.967(5) 1.979(5)	$\begin{array}{l} O(8^{ii}) - Zn(2) - O(5^{iii}) \\ O(8^{ii}) - Zn(2) - O(6^{ii}) \\ O(5) - Zn(2) - O(6^{ii}) \\ O(8^{ii}) - Zn(2) - O(4) \\ O(5^{iii}) - Zn(2) - O(4) \\ O(6^{ii}) - Zn(2) - O(4) \end{array}$	106.7(2) 115.4(2) 105.5(2) 107.4(2) 112.8(2) 109.2(2)
P(1)-O(6) P(1)-O(7 <sup>iv</sup> ) P(1)-O(4) P(1)-O(1)	1.528(5) 1.529(6) 1.538(5) 1.541(5)	$\begin{array}{l} O(6) - P(1) - O(7^{iv}) \\ O(6) - P(1) - O(4) \\ O(7^{iv}) - P(1) - O(4) \\ O(6) - P(1) - O(1) \\ O(7^{iv}) - P(1) - O(1) \\ O(4) - P(1) - O(1) \end{array}$	108.1(3) 110.8(3) 109.6(3) 110.6(3) 109.1(3) 108.7(3)
$P(2)-O(2^{v})  P(2)-O(8^{ii})  P(2)-O(3^{vi})  P(2)-O(5^{vi})$	1.532(5) 1.532(5) 1.537(5) 1.546(5)	$\begin{array}{l} O(2^{\nu}) - P(2) - O(8^{ii}) \\ O(2^{\nu}) - P(2) - O(3^{\nu i}) \\ O(8^{ii}) - P(2) - O(3^{\nu i}) \\ O(2^{\nu}) - P(2) - O(5^{\nu i}) \\ O(8^{ii}) - P(2) - O(5^{\nu i}) \\ O(3^{\nu i}) - P(2) - O(5^{\nu i}) \end{array}$	109.9(3) 111.0(3) 108.6(3) 110.9(3) 107.5(3) 108.9(3)
Li(1)-O(9 <sup>vii</sup> ) Li(1)-O(1) Li(1)-O(6 <sup>ii</sup> ) Li(1)-O(3 <sup>iv</sup> )	1.91(2) 1.972(12) 1.987(13) 2.00(2)	$O(9^{vii})$ -Li(1)-O(1) $O(9^{vii})$ -Li(1)-O(6 <sup>ii</sup> ) O(1)-Li(1)-O(6 <sup>ii</sup> ) $O(9^{vii})$ -Li(1)-O(3 <sup>iv</sup> ) O(1)-Li(1)-O(3 <sup>iv</sup> ) $O(6^{ii})$ -Li(1)-O(3 <sup>iv</sup> )	106.6(8) 115.6(7) 97.9(6) 103.3(6) 116.2(6) 117.4(8)
Li(2)-O(10) Li(2)-O(5) Li(2)-O(2) Li(2)-O(4 <sup>viii</sup> )	1.943(14) 1.971(13) 1.992(13) 1.985(13)	O(10)-Li(2)-O(5) O(10)-Li(2)-O(2) O(5)-Li(2)-O(2) O(10)-Li(2)-O(4 <sup>viii</sup> ) O(5)-Li(2)-O(4 <sup>viii</sup> ) O(2)-Li(2)-O(4 <sup>viii</sup> )	97.8(6) 118.0(6) 98.2(6) 102.9(6) 130.8(6) 109.9(6)
O(9)-H(1) O(9)-H(2) O(10)-H(3) O(10)-H(4)	0.952(5) 0.904(6) 0.972(6) 1.018(6)	Li(1 <sup>ix</sup> )-O(9)-H(1) Li(1 <sup>ix</sup> )-O(9)-H(2) H(1)-O(9)-H(2) Li(2)-O(10)-H(3) Li(2)-O(10)-H(4) H(3)-O(10)-H(4)	112.8(6) 113.3(7) 109.8(5) 111.2(6) 84.3(6) 159.4(7)

## **Synthesis**

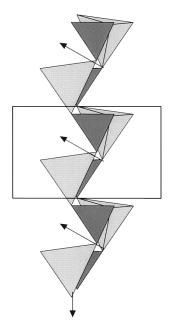
In a previous study  $\alpha$ -LiZnPO $_4$ ·H $_2$ O was found to crystallise from the system LiOH–Zn(NO $_3$ ) $_2$ -H $_3$ PO $_4$ -water. Hydrothermal syntheses from the system LiOH–Zn(CH $_3$ CO $_2$ ) $_2$ -H $_3$ PO $_4$ -water suggest that  $\alpha$ -LiZnPO $_4$ ·H $_2$ O is in fact a metastable intermediate phase that transforms to  $\beta$ -LiZnPO $_4$ ·H $_2$ O, i.e. the system approaches higher density and higher thermodynamic stability: gel  $\longrightarrow \alpha$ -LiZnPO $_4$ ·H $_2$ O  $\longrightarrow \beta$ -LiZnPO $_4$ ·H $_2$ O. The material  $\alpha$ -LiZnPO $_4$ ·H $_2$ O has only been obtained as a microcrystalline powder, whereas  $\beta$ -LiZnPO $_4$ ·H $_2$ O can be obtained as larger crystals. This is probably due to the very different timescales of nucleation and crystal growth for  $\alpha$ - and  $\beta$ -LiZnPO $_4$ ·H $_2$ O.

# Thermal investigation

Phase transformations of α- and β-LiZnPO<sub>4</sub>·H<sub>2</sub>O were investigated by hydrothermal and dry heating. Hydrothermal heating of α-LiZnPO<sub>4</sub>·H<sub>2</sub>O at ultra high pressure (80 °C, 10 kbar, 20 h) produced β-LiZnPO<sub>4</sub>·H<sub>2</sub>O, *i.e.* approaching higher density. Fig.

Table 4 Calculated oxidation states,  $V_i$ , for the structure of β-LiZn-PO<sub>4</sub>·H<sub>2</sub>O, using ref. 21

Cation	No.	$V_{\mathbf{i}}$	Co-ordination number
Phosphorus	1	5.01	4
•	2	4.97	4
Zinc	1	2.02	4
	2	2.03	4
Lithium	1	1.04	4
	2	1.02	4
Oxygen			
Framework	1	1.99	3
	2	2.00	3
	3	1.94	3
	4	1.96	3
	5	1.98	3
	6	2.01	3
	7	1.83	2
	8	1.81	2
Water	9	2.07	3
	10	1.75	3



**Fig. 5** Crystal structure of α-LiZnPO<sub>4</sub>·H<sub>2</sub>O, the b,c projection showing four-ring chains in the c direction. The arrows indicate a mechanism for a displacive phase transition of α- to β-LiZnPO<sub>4</sub>·H<sub>2</sub>O, *i.e.* the tetrahedra in the lower chain are rotated by ca. 45° and the upper chain is moved slightly downwards. The ZnO<sub>4</sub> are shown as grey and PO<sub>4</sub> as dark tetrahedra

5 shows the b,c projection of  $\alpha$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O with four-ring chains in the c direction and arrows indicate how  $\alpha$ - transforms into  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O by a displacive phase transition [compare Fig. 5 and 4(a)]. Half the chains of tetrahedra (the lower chain in Fig. 5) are distorted as indicated by the arrows, *i.e.* the tetrahedra are rotated by ca. 45°. The tetrahedra in the upper chain are all moved slightly. The mean angle between the framework tetrahedral atoms, T–O–T angles, in  $\alpha$ - and  $\beta$ -LiZnPO<sub>4</sub>·H<sub>2</sub>O are similar, 125.2 and 128.3°, respectively.

Hydrothermal heating of α-LiZnPO<sub>4</sub>·H<sub>2</sub>O at T > 100 °C was found to produce  $\delta_{\rm I}$ -LiZnPO<sub>4</sub>, and the mechanism appeared to be solution mediated. The compound β-LiZnPO<sub>4</sub>·H<sub>2</sub>O also transforms into  $\delta_{\rm I}$ -LiZnPO<sub>4</sub> upon hydrothermal treatment at >100 °C. The dry heating dehydration of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O was investigated using differential scanning calorimetry giving an enthalpy change of  $\Delta H = 43(6)$  kJ mol<sup>-1</sup>, peak value at 219(2) °C, for the irreversible phase transformation of β-LiZn-

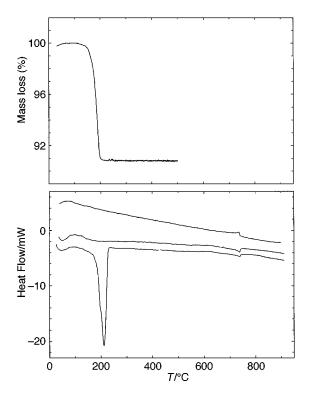


Fig. 6 Thermal investigation of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O showing the mass loss detected by thermogravimetry (upper graph) and differential scanning calorimetry; the lower and the middle DSC curves are the first and second heating and the upper DSC curve is one of the very similar cooling curves

PO<sub>4</sub>·H<sub>2</sub>O into δ<sub>1</sub>-LiZnPO<sub>4</sub>. A shoulder, at *ca.* 206 °C, was found on the DSC signal for the dehydration, Fig. 6, but the TG trace suggests a one-step process. A weak signal at 738(1) °C shows the phase transition of  $\delta_{I}$ -LiZnPO<sub>4</sub> into  $\beta$ -LiZnPO<sub>4</sub> in accordance with the equilibrium phase diagram.11 The DSC trace resembles a shift in baseline indicating that mainly a change in heat capacity is taking place. Cooling β-LiZnPO<sub>4</sub> produces  $\alpha$ -LiZnPO<sub>4</sub> [a = 17.240(2), b = 9.768(1), c = 17.106(1) Å and  $\beta = 110.90(1)^{\circ}$ , refined from powder diffraction data, 118 reflections] and the  $\alpha$ - to  $\beta$ -LiZnPO<sub>4</sub> phase transition is shown to be reversible in Fig. 6.

The zeolites  $MAlSiO_4 \cdot xH_2O$  (M = Li, Na or Ag) show a large framework flexibility as the dehydrated polymorphs may retain the ABW topology and the aluminosilicates of Na and Ag have partly dehydrated ABW polymorphs (x = 0.8 and x = 0.8 and 0.68, respectively).<sup>24</sup> Further heating of the dehydrated ABW type LiAlSiO<sub>4</sub> produces a stuffed cristobalite-type structure,<sup>24b</sup> which is similar to the phase transition of α- and β-LiZnPO<sub>4</sub>·  $H_2O$  to  $\delta_I$ -LiZnPO<sub>4</sub>.

## Conclusion

Investigation of the system LiOH-Zn(CH<sub>3</sub>CO<sub>2</sub>)<sub>2</sub>-H<sub>3</sub>PO<sub>4</sub>water, revealed that a new phase of LiZnPO<sub>4</sub>·H<sub>2</sub>O can be prepared under ambient conditions. A known polymorph of LiZnPO<sub>4</sub>·H<sub>2</sub>O (ABW structure, space group Pna2<sub>1</sub>) was found as an intermediate phase during the synthesis and transforms into β-LiZnPO<sub>4</sub>·H<sub>2</sub>O by hydrothermal heating at ultra high pressure. A mechanism for the displacive phase transition is proposed. Hydrothermal or dry heating of β-LiZnPO<sub>4</sub>·H<sub>2</sub>O at >100 °C produces  $\delta_{\Gamma}$ -LiZnPO<sub>4</sub>. The crystal structure of  $\beta$ -LiZn-PO<sub>4</sub>·H<sub>2</sub>O can be viewed as framework structures built from PO<sub>4</sub> and ZnO<sub>4</sub> tetrahedra with LiO<sub>4</sub> tetrahedra and water molecules placed in eight-ring channels running in the c direction. This compound is apparently the only member of the zeolite ABW group of materials crystallising in space group  $P2_1ab$ .

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